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EFFECT OF OXIDIZING FUELS ON AIRCRAFT FUEL SYSTEM ELAS OMERS AND SEALANTS

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20 May 1980

FINAL REPORT
NAVAIRPROPCEN Work Request N6237676WRTD107
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20	Jat fuels containing peroxides resulting from hydest fuels containing peroxides resulting from hydestes potential for damaging fuel system electores tests were conducted for 1900 hours (130 - 165) and 10 mag. peroxide/1000 gm fuel with various fuelstomers. Damage thresholds were indicated and	ic components. Immersion in J?-5 containing 1, 5, sel system sealants and
	contents were recommended.	

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#### BACKGROUND

The presence of peroxides in JP-5 fuel has been related to the failure of a neoprene diaphragm located in the high pressure pump in the TF41-A-2 engine. Examination of the fuel used by the aircraft showed the presence of peroxides which can detrimentally affect the life of the neoprene diaphragm. It is believed the peroxides result from a hydrogen treatment during the refining process which is used by several producers throughout the world. The hydrogen treatment may remove the naturally occurring anti-oxidents which normally prevent peroxides from forming in fuels refined by other methods. A few incidents of elastomer deterioration by peroxide containing aircraft fuels have previously been found in Europe as reported in reference (a).

The present investigation was conducted to ascertain any detrimental effects of peroxide containing jet fuels on various type elastomers and sealants used in aircraft fuel systems and to find a safe permissible peroxide level as a requirement in fuel specifications. This study was sponsored by the Propulsion Technology and Project Engineering Department, Naval Air Propulsion Center (NAPC), Trenton, NJ.

#### EXPERIMENTAL PROCEDURE

#### MATERIALS

The materials subjected to immersion tests are listed in Table I. The standard elastomers conformed to USAF Specification Bulletin 539. The fuel tank sealants are representative of those currently used in naval aircraft. The reticulated polyurethane foam is installed in fuel tanks as an explosion suppressant. The HMG diaphragm material consists of two layers of woven Fortisan (rayon) impregnated and coated with neoprene elastomer. The final coating thickness is approximately 2 mils on each surface. The HMG diaphragm, its neoprene coating compound and the standard neoprene were tested to obtain base line performance information.

#### TEST METHODS

- 1. The general immersion test scheme is shown in Table I.
- 2. <u>Fuels</u> a naturally occurring peroxide (hydroperoxide) concentrate, 145-185 meq. peroxide/1000 gm. JP-5 fuel, extracted from JP-5, was supplied by NAPC and used to prepare the immersion test media. The peroxide concentration was determined and monitored during the immersion tests using an analytical method developed by NAPC. The procedure involved a colorimetric potassium iodide reaction which was measured with a Bausch & Lomb Spectronic-20 Spectrophotometer and uniquely required only 0.05 to 0.5 cc of fuel sample. Peroxide concentrations were maintained within + 10% tolerance. Duplicate test specimens of each material were immersed in 200 cc of fuel.
- 3. Immersion Period a continuous immersion for 42 days was conducted with daily visual inspections and physical property measurements at 7, 14, 28, and 42 day intervals.

4. Physical Property Tests - the test methods employed were of a screening nature to detect any gross damaging effects by the oxidizing fuels. Duplicate looped configuration specimens of the elastomers and cured suslants (MIL-S-8802) were prepared from 2-inch long X 0.5 inch wide X 0.075 inch thick strips of material folded in half lengthwise and stapled midway with a stainless steel staple. The one-inch long stressed loop area above the staple was exsmined for cracks or crazing during the immersions using 15% magnification. Shore a durameter hardness was determined on the flat tail section of the loop specimen. Specimens were plied to 1/4 in. thickness for the hardness readings. Volume and weight changes were also obtained on the loop specimens. Change! sealants were applied in 1/16 in. thickness to 1 in. by 2 in. aluminum allow panels. Vigual and manual examination for changes in hardness and adhesion were performed. Weight and volume changes were also measured. All immersion tests were conducted in accordance with ASTM Method D471-72, "Standard Method of Test for Change in Properties of alastomeric Vulcanizates Resulting from Immersion in Liquids."

Change in tear strength of the polyerethane form was determined in accordance with the procedures of Specification MIL-B-83054A (USAF) and ASTM Method D1564-71 Suffix G, "Standard Methods of Testing Slab Flexible Urethane Foam."

#### TEST RESULTS AND DISCUSSION

#### 1. Neoprene Elastomers

Results of immersion tests on various elastomers and the HMG diaphragm material are shown in Table II. The neoprene diaphragm material was only lightly oxidized by 1 meq. peroxide as indicated by surface dulling after 42 days immersion. At 5 meq. peroxide, surface dulling occurred within 14 days and subsequently some light pitting. At 10 meq. peroxide heavier attack occurred and cracking of the neoprene coating initiated within 28 days immersion. Deterioration increased with continued immersion up to 42 days at which time the coating appeared weakened and could be removed by light fingernail scraping.

Looped specimens of the HMG coating compound prepared from 0.075 in. thick sheets exhibited similar surface effects; i.e., dulling and pitting, as the HMG disphragm but to a lesser degree. Very little cracking occurred. Due to the large difference in surface area to rubber volume ratios (0.002 in. film vs. 0.075 in. sheet) deterioration and cracking is most likely to occur first in the thin neoprene film. Table II hardness data shows 1 meq. peroxide produced little change. At 5 and 10 meq. peroxide a slight softening trend occurred in the swoller state indicative of some decomposition in the basic neoprene structure. In the dry state after 42 days immersion, all media caused a net hardening. Hardening with and without peroxide is caused by extraction of processing oils, waxes and antioxidants known to be present in the HMG neoprene compound. Table II data shows approximately a 5% weight loss after immersion.

The standard neoprene compound (CR) exhibited surface dulling and pitting in the JP-5/peroxide fuels similar to the HMG neoprene compound. Additionally,

a pimpled surface appeared which increased with increasing peroxide content, particularly at 5 and 10 meq. concentrations. This is believed due to an oxidation reaction. Cracking was not evident. The standard neopyene formulation differs from the HMG neoprene in polymer type, and content of fillers (carbon black, clay) and extractable oils and waxes. Reaction to the fuel media may therefore differ with regard to swelling, hardness and degree of oxidation. Hardness data show the oxidizing fuels at 5 and 10 meg. peroxide concentration caused a considerable softening trend in the swollen state compared to the JP-5 control which is indicative of a decomposition effect. This was also shown in the dry state after 42 days immersion. The less obvious softening of the HMG neoprene compound compared to the standard neoprene (CR) may be attributed to the known greater filler and extractables content of the HMG compound which would tend to mask softening due to neoprene polymer decomposition. Hardness changes in elastomers can be indicative of changes in physical properties such as tensile strength, elongation and tear strength, but measurement of physical properties is preferred.

An interesting phenomenca was found regarding the peroxide concentration of the fuel immersion media used for the HMG neoprene coated fabric and the neoprene coating compound. With all other elastomers and sealants tested herein the peroxide contents tended to decrease as immersion time progressed requiring addition of JP-5/peroxide concentrate to maintain desired peroxide levels. With the HMG materials, the 1, 5 and 10 meq. peroxide media were found to increase to approximately 3.9, 28 and 84 meg. peroxide, respectively, (in the worst cases) when allowed to escalate. The maximum increases usually occurred during the 20 to 30 day immersion time perod. The control JP-5 fuel also showed an increase from 0.16 to approximately 0.3 meq. peroxide. Further, JP-5/peroxide fuels without elastomer samples aged at 130°F for 42 days all showed gradual decreases in peroxide content. In general, media containing HMG elastomer samples required almost daily monitoring of peroxide content. Evidently the HMG neoprene elastomer produced a catalytic effect and peroxide content escalation. Certain metals, e.g., copper, are known to have catalytic effect on fuels. Possibly contaminant metallic elements in the elastomer compounding ingredients; i.e., clays, arbon black, oils, waxes or resideal copper catalyst used to manufacture neoprene-polymer may be responsible.

#### 2. Nitrile Elastomers

Table II data shows that the nitrile standard elastomers were visibly unaffected except for development of a shiny surface after 42 days immersion. This effect is probably due to light surface oxidation. Cracking was not evident. Hardness data indicate the low nitrile elastomer (NBR-L) was unaffected by 1 mag. peroxide and showed a slight softening trend in the swollen state at 5 and 10 mag. peroxide contents compared to the control. In the dry state hardness did not differ significantly from the controls. Hardness changes of the high nitrile elastomer (NBR-H) in all JP-5/peroxide fuels were equivalent to the controls indicating no ill effects. In the dry state a very minor hardening trend mostly at 5 and 10 mag. peroxide contents appeared. It should be noted the NBR-H elastomer is much more fuel resistant than the NBR-L elastomer and has considerably less volume swell and fuel absorption as shown in Table II.

Nitrile seals for aircraft tend to be fabricated from medium to high nitrile elastomers rather than NBR-L. Components such as nitrile coated disphragms may be more susceptible to oxidative physical changes in view of the high surface area to volume ratio. The nitrile and neoprone elastomers are known to be susceptible to oxidation, consequently antioxidants are generally incorporated in their manufacture. Effectiveness of the antioxidants would depend upon the degree of their fuel extractability. From a functional standpoint, size and shape of an elastomeric component and its operating stresses may also influence the effects of an oxidizing medium.

# 3. Fluorocarbon and Fluorosilicone Elastomers

Table II data indicates the standard fluorocarbon elastomer (F-A) immersed at a higher temperature, 165°F, in 10 meq. peroxide fuel exhibited no significant visible or hardness changes compared to the control. The fluorosilicone elastomer (FS) behaved similarly except for a minor hardening trend. Note that peroxides are used to vulcanize the FS elastomer. Both the F-A and FS elastomers are known to be highly fuel and oxidation resistant.

immersion of these elastomers was first attempted at 200°F, but the 10 meq. peroxide content decreased so rapidly it could not be practically maintained; 165°F was the next highest temperature at which the 10 meq. peroxide concentration could be practically maintained.

#### 4. Polysulfide Cured Sealants (MTL-S-8802)

Test results on the elastomeric MIL-S-8802 sealents are given in Table III. After 42 days in 10 meq. peroxide fuel the chromate cured PR-1422B-2 scalant showed significant softening compared to the control, 56 vs. 65 Shore A, respectively, in the dry state which is indicative of a scalant degradation effect. A comparative manual tear strength examination also indicated some loss in strangth due to the peroxide fuel exposure. The manganese dioxide cured PR1440B-2 after 42 days immersion also showed a net softening compared to the control, 51 vs. 61 Shore A, respectively, in the dry state. The peroxide-fuel also caused a tacky surface and loss in tear strength. Note the PR1440B-2 control specimen increased in hardness after impersion (55 to 61 Shore A) probably due to entraction of plasticizer which is used with the manganese dioxide curing agent. The sample exposed to peroxide-fuel did not follow the control's behavior and softened instead (51 Shore A) indicating a degradation effect. Subsequently, one specimen of PR1440B-2 was subjected to additional immersion in 10 mag. peroxide fuel for a total of 56 days. Hardness decreased further to 45 Shore A and the sealant surface further degraded to a semi-fluid condition along with other changes indicated in Table III. This degradation is attributed to a reversion effect caused by polymer chain and/or crosslink scission. Polysulfide polymers are susceptible to attack by acid catalyzed hydrolysis at the backbone formal linkage. Only trace amounts of water are needed. Sulfur, mercaptans and strong bases under proper conditions may also cause polymer degradation at the disulfide linkage. This mode is unlikely in view of the good performance of the control sealant. Since periodic additions of fresh peroxide fuel concentrate were required to maintain

10 maq. peroxide it was of interest to ascertain if the peroxide, or strictly hydroperoxide degradation products could result in acidic build-up in the immersion fluids. Acid number analysis, performed by NAPC, on the PR1440B-2 immersion media were as follows:

- a. JP-5 control 0.007 after 42-day immersion
- b. JP-5/10 meq. peroxide 0.028 0 days immersion
- c. JP-5/10 meq. peroxide 0.043 after 42-day immersion (JP-5 specification limit MIL-T-5624 0.015 (max.))

The data indicates acidic constituents of the peroxide fuel exceeded both the JP-5 control value and specification limits and may have contributed to seal-ant degradation. Preparation of the original JP-5/peroxide concentrate may also have caused some acidic buildup. Another polysulfide degradation route is by direct oxidation at the formal linkage but this usually occurs at temperatures around 300 T. The polysulfides are generally considered to have good oxidation resistance at temperatures below 200 F. They are also cured by inorganic and organic oxidizing agents. From the available data it is uncertain whether catalyzed oxidation or acid catalyzed hydrolysis at the formal linkage is the primary degradation process. The latter mechanism is favored. From a practical standpoint, assessment of the acid number of peroxide fuels in aircraft fuel tanks, particularly where stagnant fuel resides, would be of interest to further determine the potential for sealant degradation.

### 5. Non-Curing Channel Sealants

Table III test data on the polysulfide, cyanosilicone and fluorosilicone mastic type sealants indicate they were not significantly affected by the 10 meq. peroxide fuel immersion. From a cursory manual examination it appeared the polysulfide sealant immersed in peroxide fuel had undergone a slight change by becoming somewhat softer and cackier than the control after 42 days immersion. This behavior may possibly be analagous to that found with the cured type polysulfide sealants. It should be noted that channel sealants are manufactured with a high degree of tack to achieve adhesion.

#### 6. Polyursthane Foam - Fuel Tank Baffle (MIL-B-83054A)

Results of immersion tests are given in Table IV. Tear strength determinations indicate the 10 meq. peroxide fuel did not produce any adverse effects in comparison with the control fuel. Both fuels caused about a 15% decrease in tear strength.

Although not indicated by the results herein, it should be noted the polyurethane foam is susceptible to degradation by hydrolysis which can be accelerated by an acidic medium. In this regard, high acid number peroxide fuels would be undesirable.

#### CONCLUSIONS

From the 1000 hour screening immersion tests with JP-5/peroxide fuels conducted herein, the following is concluded:

1. Serious damage thresholds for the HMG disphragm are estimated as follows:

1 meq. peroxide 1000 hours

5 meq. peroxide 670 - 1000 hours

10 meq. peroxide 35 - 670 hours

Dynamic operating conditions can be expected to shorten these damage thresholds.

- 2. The standard neoprene elastomer (CR) behaved similarly to the HMG neoprene material.
- 3. In general, thin elastomeric films (e.g., diaphragms) would be more readily deteriorated by oxidizing fuel than comparatively thick elastomer items (e.g., 0-rings).
- 4. Low and high nitrile elastomers (NBR-L, NBR-H) were relatively unaffected by 1 meq. peroxide fuel during the 1000 hour exposures. At 5 and 10 meq. peroxide levels the NBR-L elastomer is susceptible to a very mild oxidative attack manifested by minor softening. The NBR-H appears more resistant and only showed a tendency to slightly harden after the 1000 hour exposure. In general, the observed effects are not believed indicative of major changes in the physical strength properties of the NBR elastomers.
- 5. Fluorocarbon and fluorosilicono elastomers are highly resistant to 1-10 meq. peroxide fuels and remained essentially unchanged. Higher levels or peroxide may be well tolerated.
- 6. MIL-S-8802 elastomeric polysulfide sealants, particularly manganese dioxide cured type, are prone to severe degradation in 10 meq. peroxide fuel (1000-1350 hrs.) where the fuel acid number exceeds normal limits. It is uncertain whether a catalyzed direct oxidation or an acidic hydrolysis mechanism is responsible. The latter mode is believed more likely to occur. Elevated fuel acid numbers are believed due to peroxide decompostion by-products possibly present in the original prepared JF-5/peroxide concentrate and additionally accumulated in the stagnant immersion media during the test phase. Data on the effects of lower peroxide concentrations and acid numbers is needed to further assess damage potential with MIL-S-8802 sealants. These tank sealants are intended to provide very long service and are difficult to repair.
- 7. Polysulfide, cyanosilicone and fluorosilicone non-curing type channel sealants exhibited good resistance to 10 meq. peroxide fuel. The polysulfide sealant was slightly softened but not enough to impair its functioning.
- 8. The MIL-B-83054A Type I polyurethane foam (tank baffle material) was not significantly affected by 10 meq. poroxide fuel during the 1000 hour exposure period.

- 9. In general, a 1 to 2 maq. peroxide limit for JP fuels should provide a good margin of safety against serious deterioration of most fuel system clastomers and sealants.
- 10. Further information on the relationship between peroxide content and scid numbers of fuels would be useful.

# RECOMMENDATIONS

- 1. The maximum allowable peroxide content of JP fuels be kept in the range of 1 2 meq. peroxide/1000 gm. fuel.
- 2. Conduct immersion studies with MIL-S-8802 sealants to determine the effects of peroxide concentrations below 10 meq. and of fuel acid number variation.
- 3. Determine the potential for acid number build-up in fuels both in storage and in operating aircraft fuel tanks.
- 4. Develop an alternate coating for the HMG diaphragm from more fuel and oxidation resistant low temperature elastomers such as fluorocarbon (Viton GLT), fluorosilicone or fluorophosphonitrilic (PNF).

#### REFERENCES

(a) "Fuel Related Problems in Engine Fuel Systems," Love B. E., et al; SAE, Aeronautic Space Engineering and Manufacturing Meeting October 3-7, 1966

#### ACKNOWLEDGHENT

The assistance of the Materials Synthesis and Characterization Section in performing the final peroxide content analyses is greatly appreciated.

TABLE I

MATERIALS AND INMERSION TEST SCHEME

		roxide			
Materials	0	1./1000 1	<u>sa. fu</u> 5	10	Immersion Or
BB 541 AB 48				10	Temperature, OF
ELASTOGERS					
HMG Diaphragm	x	x	x	x	130
Seoprene-HSG Coating Elastower	x	x	x	x	130
Neoprene-Standard CR	x	x	x	x	130
Low Sitrile-Standard NBR-L	x	x	X	x	130
High Mitrile-Standard NER-H	X	x	x	x	130
Fluorocarbon-Standard F-A(Viton A)	x	-	-	x	165
Fluorosilicone-Standard FS	X	-	-	x	165
Polysülfide Sealants - MIL-S-8802					
Cironate Cure	X	-	-	X	130
Manganese Dioxide Cure	X	•	-	X	130
Polyurethene Fosm - MIL-B-83054A, Type I	x	-	-	x	130
CHANNEL SEAL NTS					
Polysulfide - PR-702	X	-	-	X	130
Cyanosilicone - G250	x	-	-	x	130
Fluorosilicone - DC 94-011	×	•	· _	X	130

NADC-80076-60 TABLE II IMMERSION OF ELASTOMERS IN JP-5/PEROXIDE FUELS

ELASTOMER			HMG Dia rene Co					e Coating 1G Diaph:		Neoprene Standard (CR)				
JP-5 Peroxide Content Meq./1000 gm. fuel		JP-5 ( Contr		5	10	JP-5 Conti	ol 1	5	10	JP-5 Cont:	rol 1	5	:	
Immersion Temperature				130°F			:	130 <sup>0</sup> F				130°F	1	
AFTER DMERSION -	DAYS							<b>_</b>					1	
Volume Change, %													:	
	7	41.6	35.0	23.6			41.0		41.5	62.5	64.0			
	42	30.0	30.2	19.3		40.3	37.5	31.0	30.0	54.6	57.0		į	
dried 48 hrs.@ 160°F	42	-0.3	-2.0	-6.6	-5 <b>.8</b>	-7.4	-7.3	-7.5	-7.3	1.5	2.0	2.8	•	
Veight.														
Change, Z	7	9.2	9.4	10.0	10.2	18.3	18.7	18.7	18.9	33.2	34.0	35.2	4	
_	42	9.0	8.6	8.6		16.1	16.3	14.8	14.4	29.0	30.6	33.0	i	
dried 48 hrs.@ 160°F	42	-4.9	-5.6	-4.6	-4.5	-4.8	-4.6	-4.6	-4.7	-0.5	-0.7	-0.5	;	
Hardness, pts.	0	NA (2)				72	? <b>2</b>	72	72	72	72	72	1 2	
(Shore A)	7					58	58	57	57	50	50	47	1	
	14					58	58	57	55	50	48	44	4	
	28					57	57	54	53	48	45	38	1	
	42					57	57	54	53	45	43	36	- 1	
dried 48 hrs.@ 160°F	42	-				77	77	78	80	64	63	59	į	
Observations	7	OK	OK	OK	OK	OK	OK	OK	OK	OK	OK	OK	1	
1	14	OK	OK	D	D	OK	OK	OK	D	OK	OK	OK		
	28	OK	OK .	D	C,P,D (5)	OK	OK	D, few P	D,P	OK	OK	D,PM	D;	
	42	OK	D (3)	D,P <sup>(4</sup>	incr. D.P.	C OK	few P		D,P	ОК	slt.D	D,PM,P	11	
dried 46 hrs.@ 160°F	42	slt.D	D	D,P	D,P,C <sup>(6)</sup>	few P	few P	D,P	D,P (7) few C	slt.PM (8)	slt.D PM	D,PM,P	D,	

NOTES: (1) no added peroxide; 0.16 meq./1000 gm naturally present

(2) not applicable

(3) D - dulled (4) P - pits (5) C - cracks

(6) coating weakened, patches removable

(7) surface harlened (8) PM - pimpled surface (9) SS - shiny surface

	Neop: Standa	rene rd (CR)_			Low N	itrile (NBR-)	<u>L)</u>	H St	igh Nit	rile (NBR-H)		Fluoroca Standard JP-5	rton (F-A)	Fluorosi Standard JP-5 Control	(FS)
5				JP-5		5	10	JP-5 Contro	o 1	5	10	Control	Control 10		10 50F
tro	1 1	5	10	Contr								16:	5°F	10	
		130 <sup>0</sup> F			13	0°F				0°F			-		
5 16	64.0 57.0 2.0	70.5 63.5 2.8	67.5 63.0 2.5	38.5 32.0 -0.1	40.0 32.6 -0.5	41.0 32.4 2.0	40.5 31.5 3.5	17.5 17.8 4.2	15.3 13.6 3.9	15.2 13.5 4.5	15.3 13.9 6.1	1.6 2.3 1.5	2.1 4.1 2.4	8.9 4.5 1.4	9.3 5.5 0.4
2 0 5	34.0 30.6	35.2 33.0	36.2 33.0	24.8 20.5	25.8 21.7	25.7 22.6 1.4	27.2 22.8 3.0	9.2 11.0 1.9	9.4 6.6 2.1	9.7 9.3 3.1	10.1 9.8 4.2	1.1 0.6	1.4 1.4 1.1	3.2 2.2 -0.2	4.0 2.6 -0.6
5	-0.7 72 50 48 45 43	-0.5 72 47 44 38 36	-0.4 72 45 41 33 33	-1.7 70 55 54 55 55	-1.2 70 55 55 53 55	70 56 53 50 51	70 53 50 49 51 65	70 63 63 62 63 66	70 63 63 62 63 67	70 62 62 62 62 62 69	70 63 63 62 62 70	73 70 69 69 70 72	73 69 68 67 70 72	67 59 58 58 62 64	67 58 59 61 64 67
and the second of the second of the second of	63 OK OK OK	OK OK D, PM	53 OK D D,FM,P	67 OK OK OK OK	67 OK OK OK OK	65 OK OK OK OK	OK OK OK OK	OK OK OK OK	OK OK OK	OK OK OK	OK OK OK	OK OK OK	OK OK OK	OK OK OK	or or or
عنوسا	slt.D slt.D	D,PM,P	iner. D,PM,P D,PM,P	slt.D	OK OK slt.SS	ok ss	OK >SS	D	OK	OK, SS	0 <b>k,</b> >\$\$	OK	OK .	OK.	sit. D

TABLE III

# IMMERSION OF FUEL TANK SEALANTS IN JP-5/PEROXIDE FUELS

		ELASTOMERIC, CURING TYPE - MIL-S-8802									
SEALANT		PR-14 Polysulfide-	22B-2 Chromate Cure	PR-144CB-2 Polysulfide-Manganese Dioxide Cure							
P-5 Peroxide Content eq/1000 gm. fuel		JP-5(1) Control	10	JP-5 Control	10	JP- Cor					
mmersion Temperature		1	30 <sup>0</sup> F		130°F						
FTER IMMERSION -	DAYS	٠.									
_				_	••						
olume Change, 7	-	-	6.?	0.4	0.7	2					
	7	6.5 2.4	3.3	~0.3	-1.3	2					
ried 48 hrs @ 160°F	42 42	-2.1	-2.6	-7.7	-7.0	-2					
odeht Chance 7	7	1.3	1.8	-1.2	-1.0	1					
eight Change, %	42	1.0	1.7	-1.0	-0.8	1					
ried 48 hrs @ 160°F	42	-2.4	-1.7	-4.9	-4.1	-1					
ar <u>dness</u>	0	67	67	55	55	N					
Shore A)	7	61	58	54	<b>53</b> .	•					
	14	58	54	53	51	•					
	28	57	56	53	52	•					
	42	58	53	54	43/35(4)	•					
ried 48 hrs @ 160°F	42	65	56	61	51/45(4)	•					
bservations	7	OK	OK	OK	OK	0					
	14	OK	OK	OK	QK	0					
	28	OK	OK	OK	OK	0					
• _	42	OK	OX	OK	tacky	sli					
ried 48 hrs @ 160°F	42	OK	some loss	OK	tacky; loss	end bus					
			in tear (3)		in tear	4:The					
			strength		strength						
					very tacky; surface						
					e 9 a 4 4						
					gumny, weak	(4)					
dhesion to Aluminum		<sub>RA</sub> (2)	NA	AK	AM	ď					

(1) no added peroxide; 0.16 meq/1000 gm naturally present
(2) not applicable
(3) monual examination
(4) after 56 days immersion NOTES:

PR-792 Polysulfi	lde		-250 silicone	DC 94011 Fluorosilicone			
JP-5 Control	10	JP-5 Control	10	JP-5 Control	10		
130°F		j.	30 <sup>c</sup> r	130°F			
- 2.5	- 1.8	- 1.7	- 2.2	- 3.2	3.1		
2.5 -2.8	3.6 -2.2	3.3 -2.2	4.2 -1.6	3.0 -2.1	3.0 -2.1		
1.6 1.7 -1.1	1.6 2.4 -0.5	1.2 2.0 -1.1	1.5 2.5 -1.0	1.1 0.9 -1.8	1.0 9.9 -1.8		
VA (2)	NA.	NA.	· NA	ra -	ra 		
•	•	-	-	-			
-	-	-	- -	-	-		
OK OK	OK OK OK	OK OK	OK OK OK	OK OK OK	OK OK		
X ightly harder I less tack	OK slightly softer and tackier than control	OK OK	OK OK, slightly yellowed	OIK OIK	OK.		

OK

OK

CK

17

OK

TABLE IV

IMMERSION OF FUEL TANK BAFFLE MATERIAL IN JP-5/PEROXIDE PUELS

JP-5 Peroxide Control meq/1000 gm. fuel	enf.	Polyurethane Foam  MIL-B-83054A, Type I  JF-5 (1) 10  Control							
			<del> </del>						
Inmersion Temperature	<b>e</b>	130°F	<del></del>						
After Immersion-	Days								
Tear Strength, 1bs/i	n.								
•	0	7.2	7.2						
	7	5.4	5.6						
	42	6.4	6.0						
Dried 48 hrs @ 1600F	42	5.7	5.9						
Observations	42	OK	OK						

Notes (1) No added peroxide: 0.16 meq/1000gm naturally present.

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